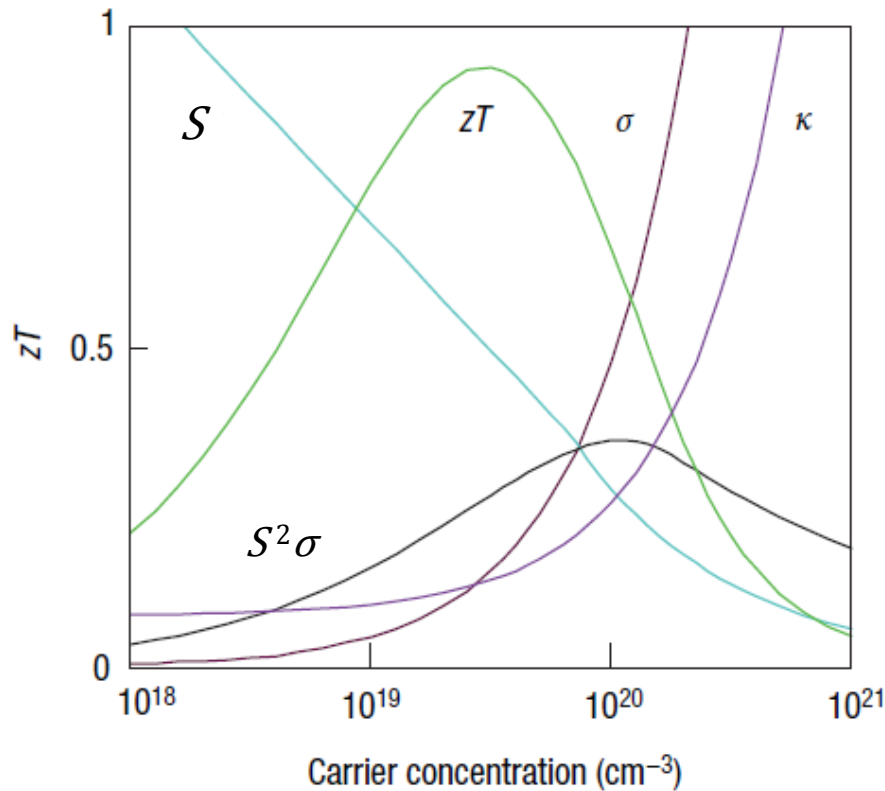


New n-Type Thermoelectric Oxides with the Perovskite Structure

Luke Daniels

UK Thermoelectric Network Meeting
Edinburgh 2018

Thermoelectrics



- m*^{*} – carrier effective mass
- n* – carrier concentration
- μ* – carrier mobility
- L* – Lorenz factor ($2.4 \times 10^{-8} \text{ J}^2 \text{ K}^{-2} \text{ C}^{-2}$)
- C_v* – heat capacity
- l_{ph}* – phonon mean free path
- v_s* – mean velocity of sound (phonons)

Figure of merit:

$$ZT = \frac{S^2 \sigma}{\kappa} T$$

Seebeck (for metals or semiconductors):

$$S = \frac{8\pi^2 k_B^2}{3eh^2} m^* T \left(\frac{\pi}{3n} \right)^{2/3}$$

Electronic conductivity:

$$\sigma = ne\mu$$

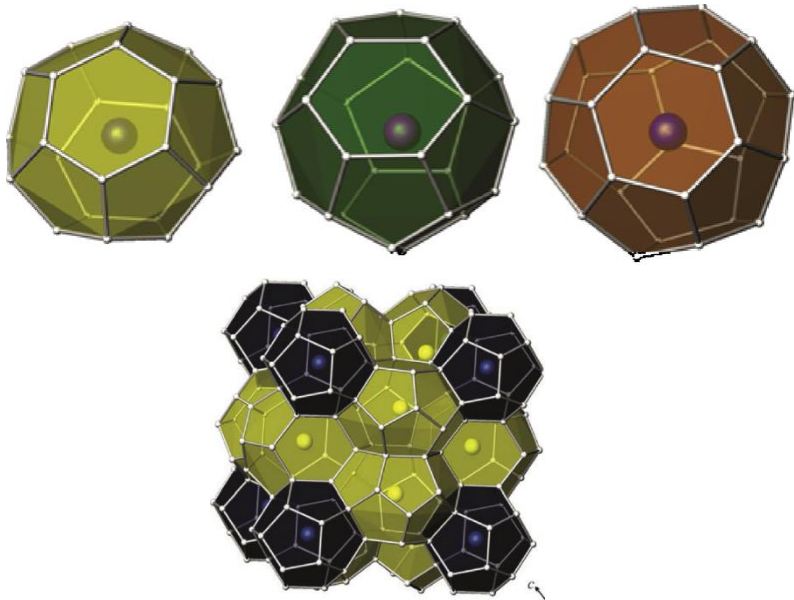
Thermal conductivity:

$$\kappa_{total} = \kappa_{elec} + \kappa_{latt}$$

where:

$$\kappa_{elec} = L\sigma T \quad \text{and} \quad \kappa_{latt} = \frac{1}{3} C_v l_{ph} v_s$$

Phonon glass electron crystals



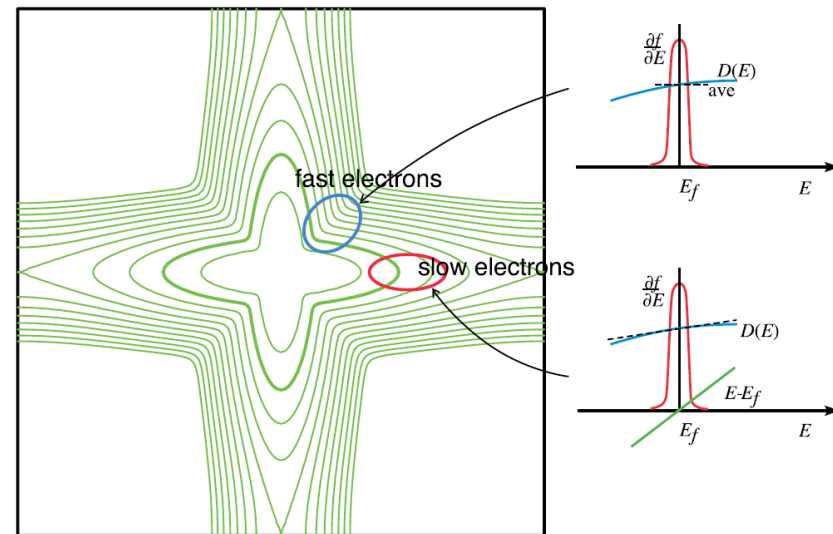
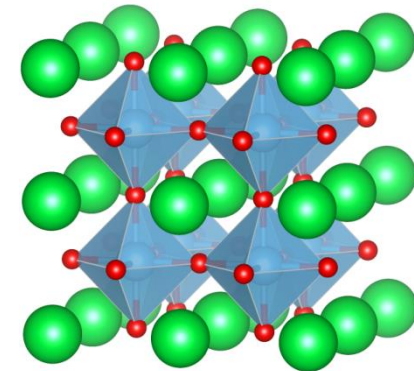
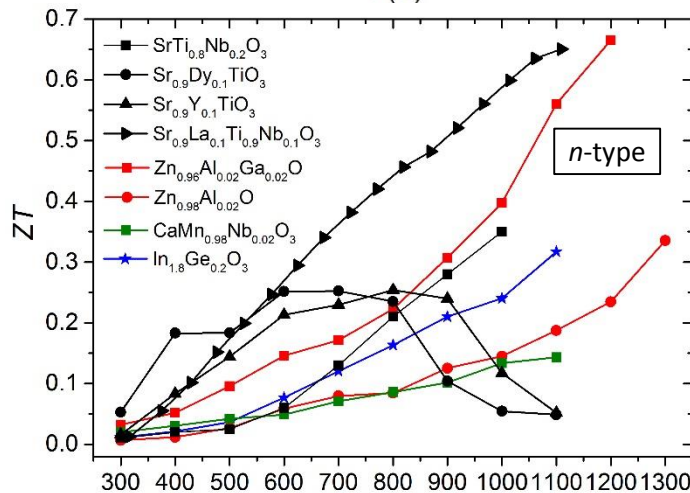
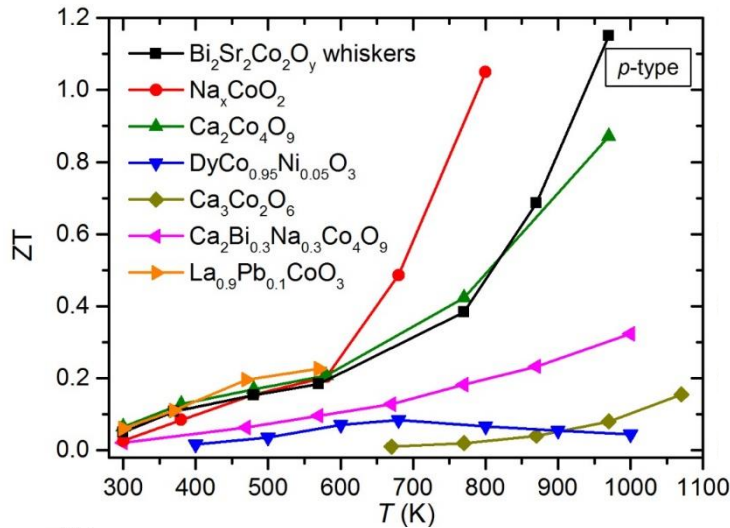
- Clathrates and skutterudites with framework structures containing guest cations.
- Loosely-bound “rattler” guest atoms lead to reduction of phonon velocities and therefore thermal conductivity.
- This does not inhibit electronic transport which is mediated through covalent framework.

Reducing intrinsic thermal conductivity:

- Increased structural complexity
 - Large unit cell
 - High molecular weight
 - Layering/superlattices
- Disorder within the unit cell
 - Mass/strain disorder through alloying
 - Rattler effects
- Chemical bonding
 - Strong anharmonicity
 - Soft chemical bonding

Oxides

- Titanates have unique electronic structures with multiple carriers resulting from high symmetry of crystal structure



Titanates

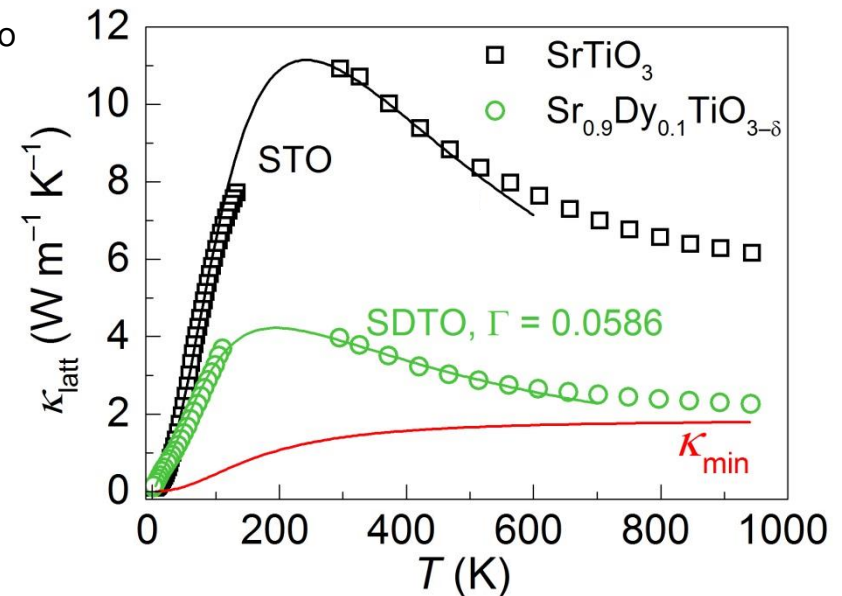
Lattice contribution dominant in oxides and is proportional to phonon mean free path (l_{ph}):

$$\kappa_{latt} = \frac{1}{3} C_V l_{ph} v_s$$

Phonon mean free path is directly proportional to phonon relaxation time (τ)

Matthiessen's rule for scattering mechanism contributions to total thermal resistance:

$$\tau_{ph}^{-1} = \tau_B^{-1} + \tau_{pd}^{-1} + \tau_U^{-1} + \tau_{LI}^{-1} + \tau_{res}^{-1}$$



Material	Γ_{MF}	$\kappa_{300 K}$
$Sr_{0.9}Y_{0.1}TiO_3$	1.93×10^{-5}	5.50
$Sr_{0.9}Dy_{0.1}TiO_3$	5.58×10^{-2}	4.50
$Sr_{1-x}La_{0.67x} \square_{0.33x}TiO_3, x=0.8$	0.404	2.50
$La_{0.5}Na_{0.5}TiO_3$	0.513	?

Phonon relaxation time due to point defect scattering:

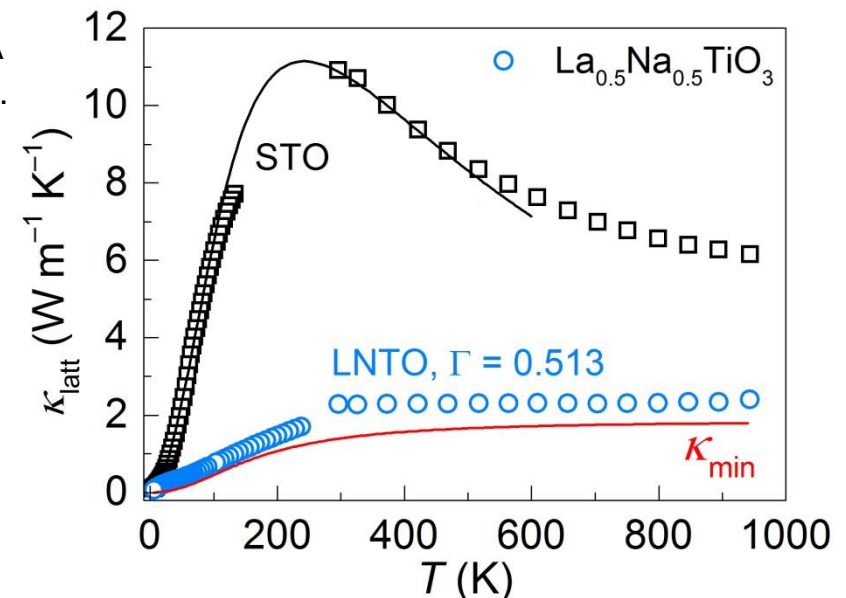
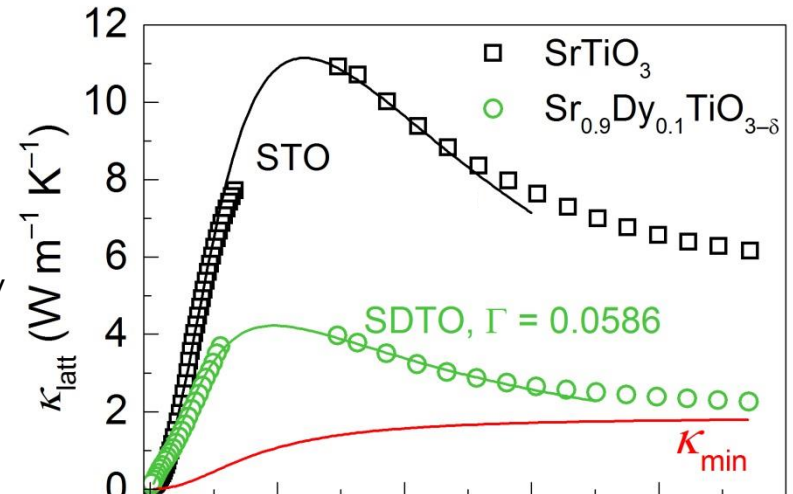
$$\tau_{pd}^{-1} = \frac{V\omega^4}{4\pi v_p^2 v_g} \left(\underbrace{\sum_i f_i \left(1 - \frac{m_i}{\bar{m}}\right)^2}_{\Gamma_{MF}} + \sum_i f_i \underbrace{\left(1 - \frac{r_i}{\bar{r}}\right)^2}_{\Gamma_{SF}} \right)$$

Point defect scattering

Cation substitution:

- STO and SDTO both display phonon-crystal characteristics
 - typical T^{-1} dependence of κ at high temperatures
- Experimental data of STO and SDTO reproduced well by modified Callaway model for crystalline materials
- Γ_{MF} of LNTO contributes 99.9% of total Γ parameter
 - Γ_{SF} contribution is negligible
- Introduction of high-mass contrast “disorder” through A site substitution reduces κ by 80% at room temperature.

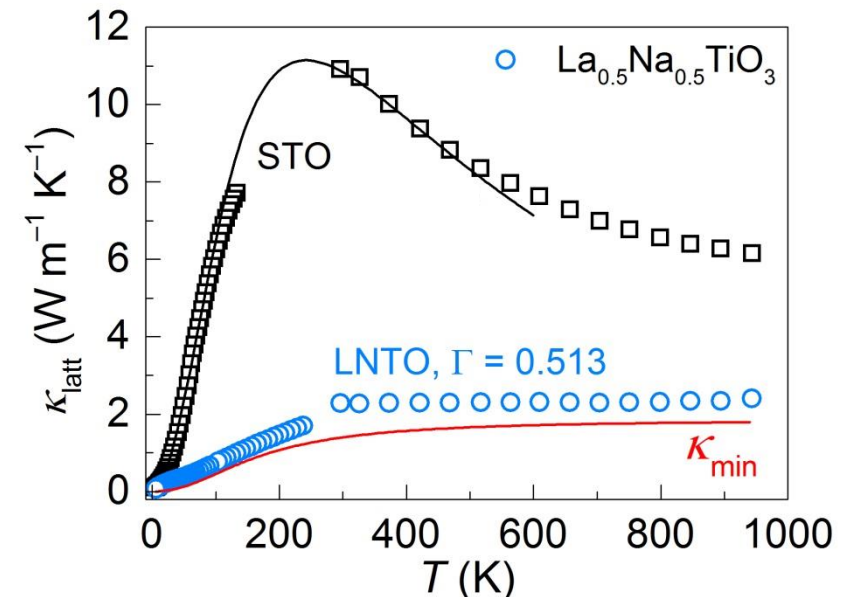
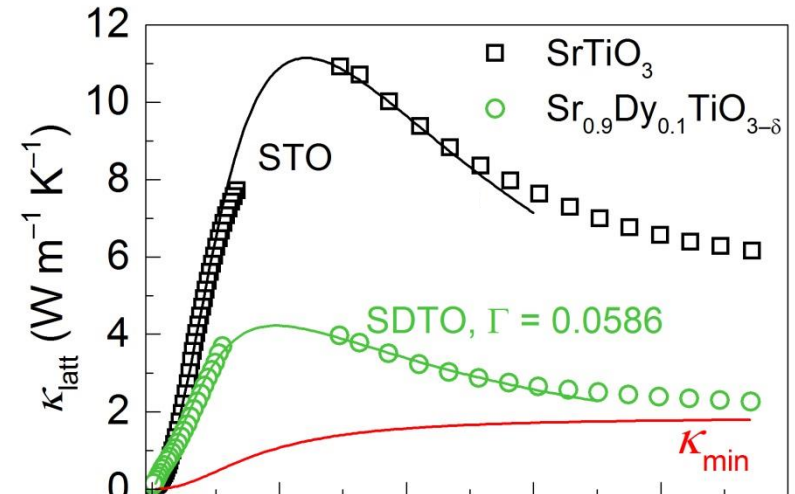
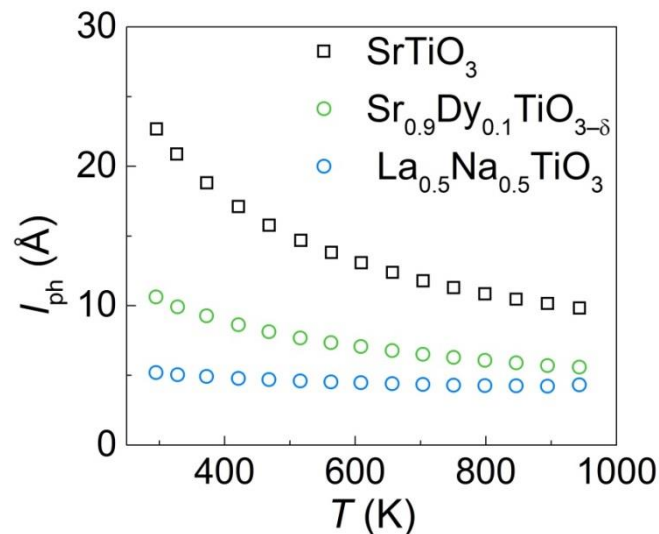
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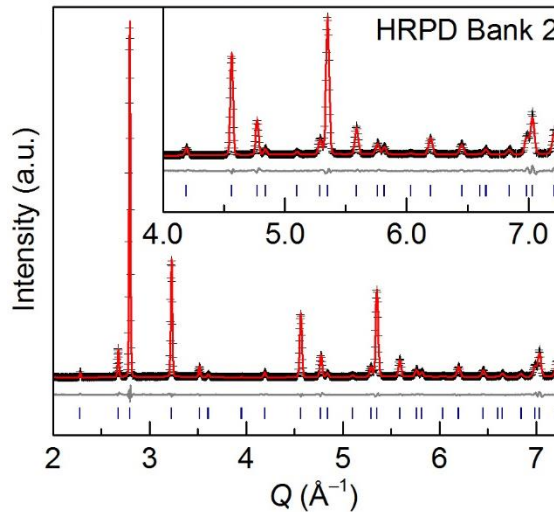
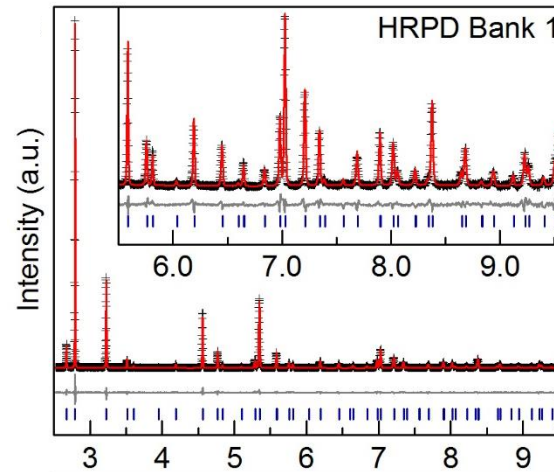
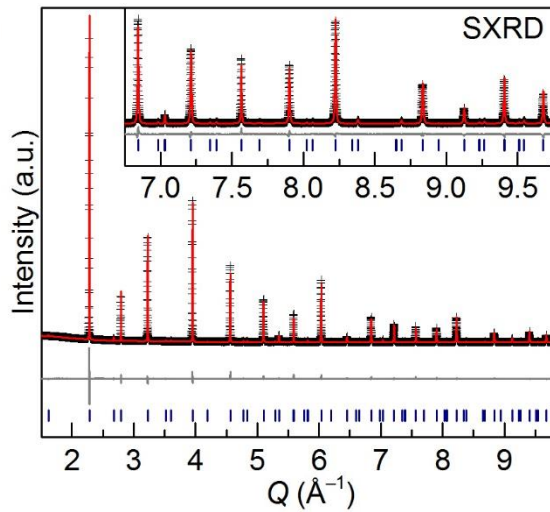
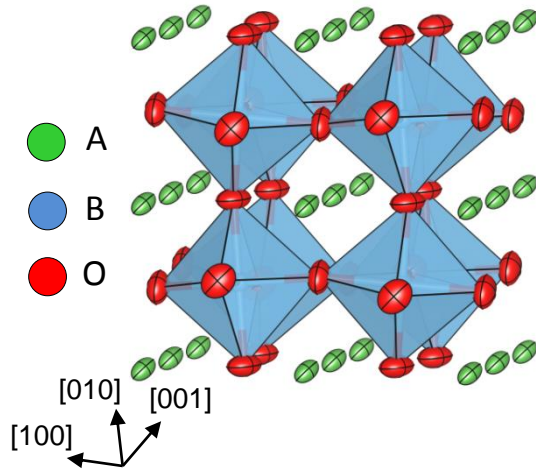
Phonon mean free path

Cation substitution:

- Constant κ of LNTO above room temperature
- Close comparison against theoretical minimum (κ_{\min}) given by Cahill model for disordered solids
- Constant phonon MFP as function of T for LNTO due to vibrations localised over interatomic distances



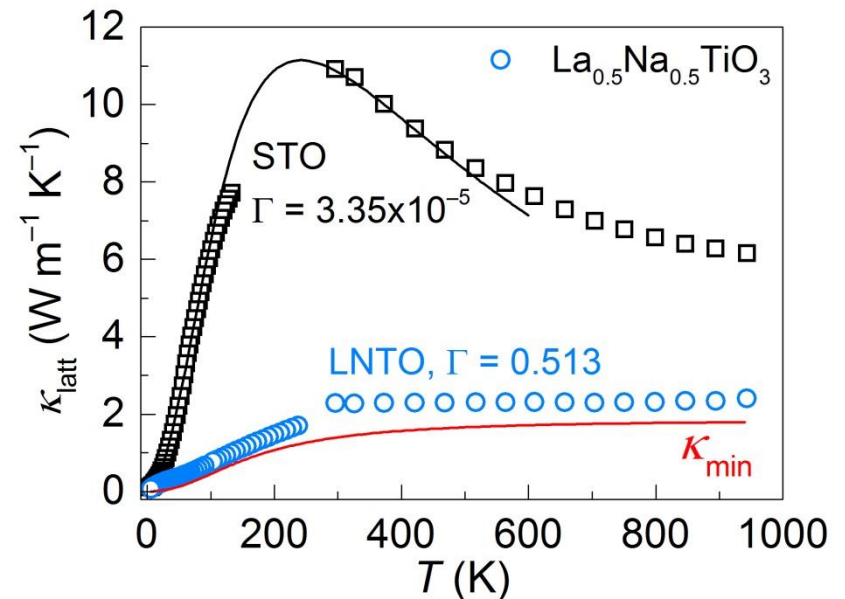
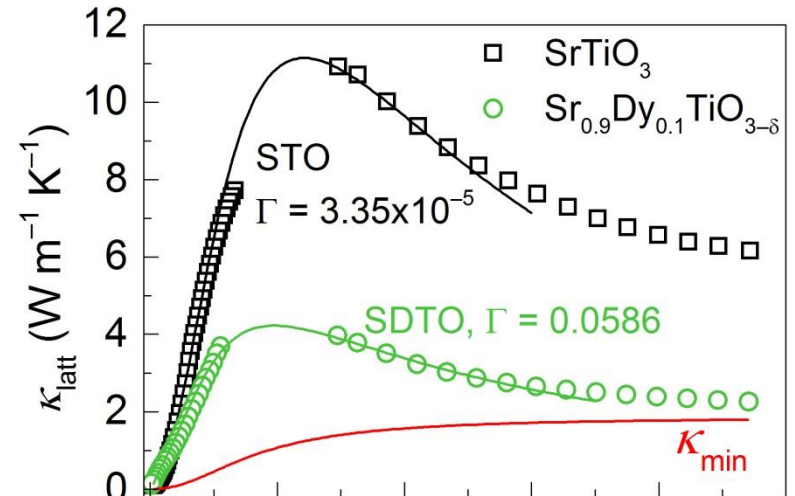
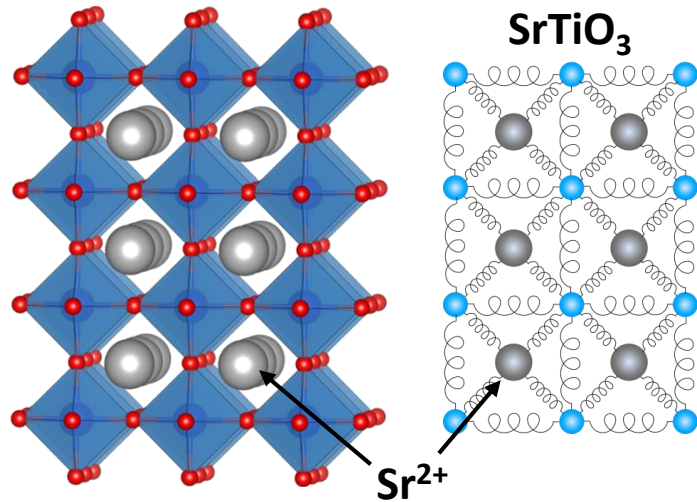
A site disorder



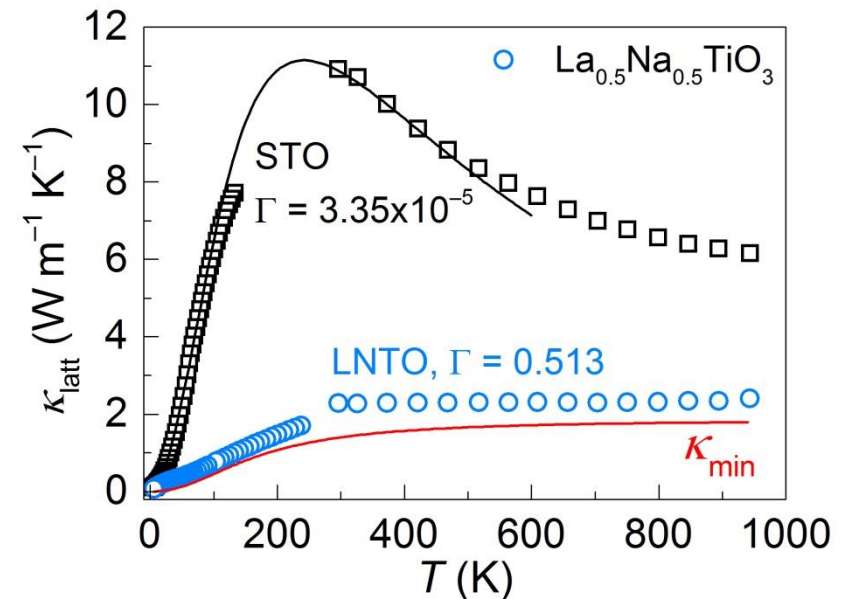
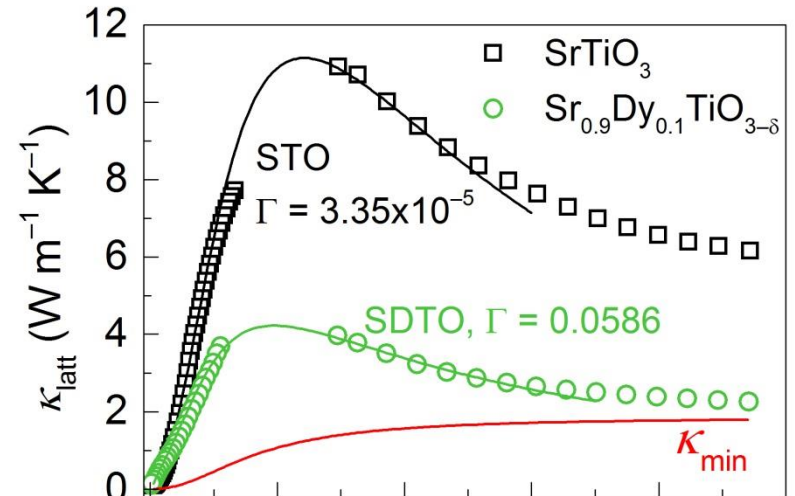
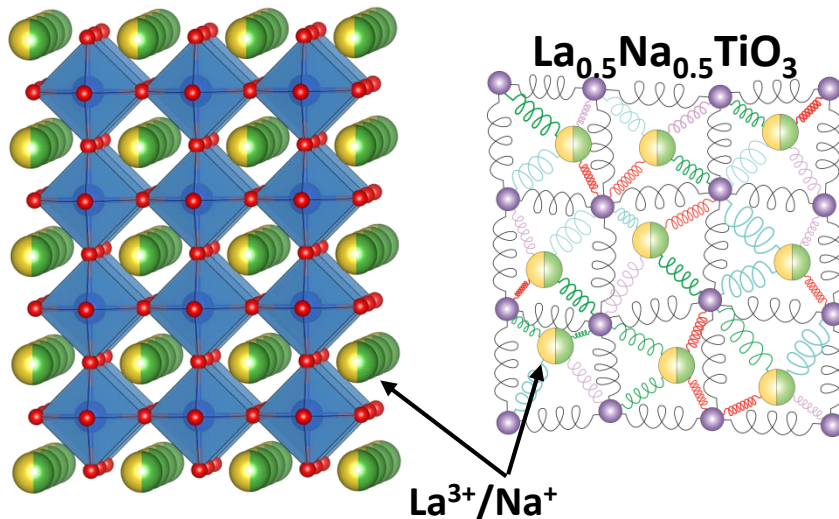
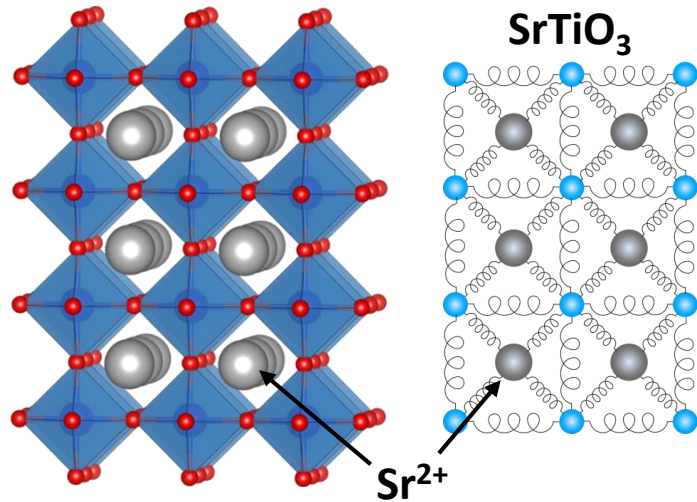
Cation disorder:

- Despite glassy κ , material is perfectly crystalline
- A site cations disordered across the site
- Glassy behaviour of κ arises from random distribution of high-mass contrast cations
- Yields phonon-glass crystalline properties

Phonon glass $La_{0.5}Na_{0.5}TiO_3$



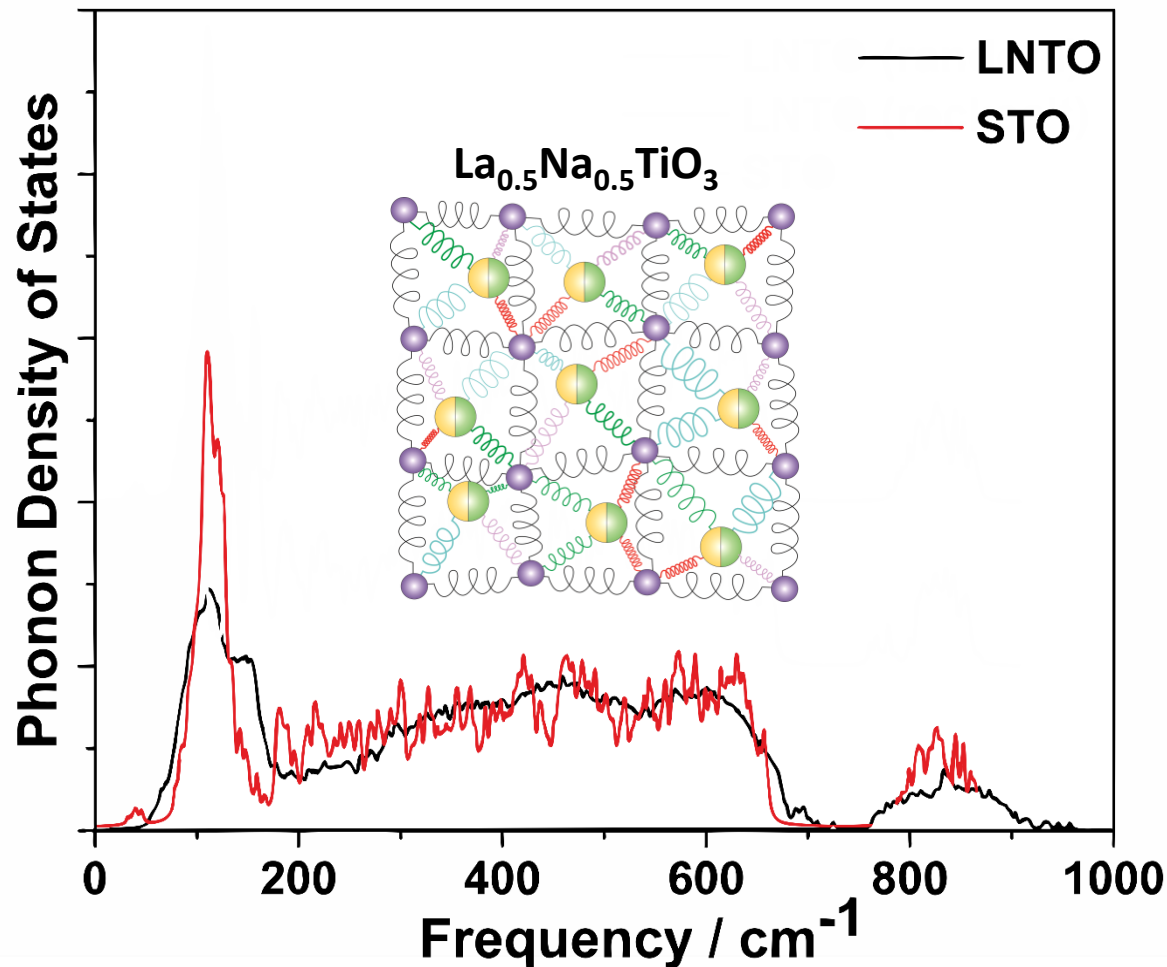
Phonon glass $La_{0.5}Na_{0.5}TiO_3$



Static DFT calculations

Phonon density of states:

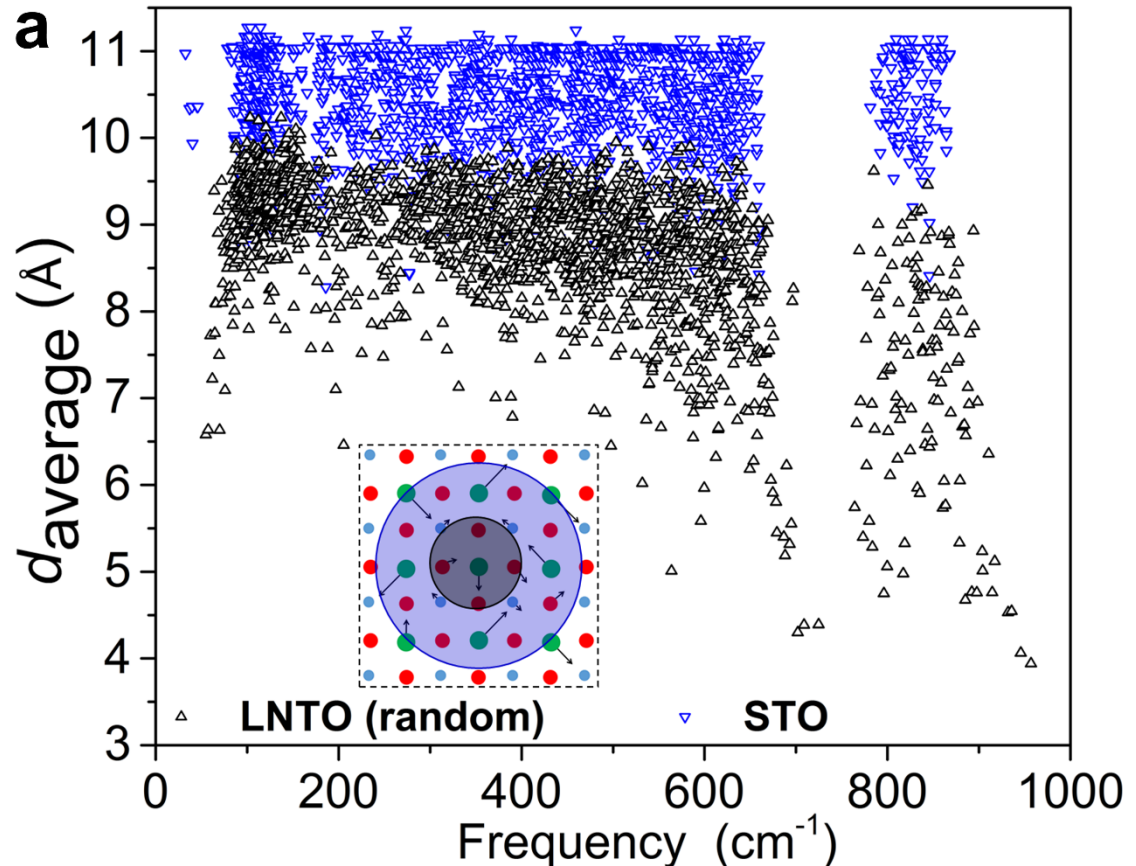
- Low frequency region ($<300\text{ cm}^{-1}$) contributes heavily to thermal conductivity
- Region well defined (narrow) for SrTiO_3
- Resonance at $\sim 110\text{ cm}^{-1}$ agrees well with value used in Debye-Callaway model used to describe κ
- A site cation disorder and high mass contrast results in broadened low-frequency region
- Flattening of phonon dispersions similar to clathrates and skutterudites

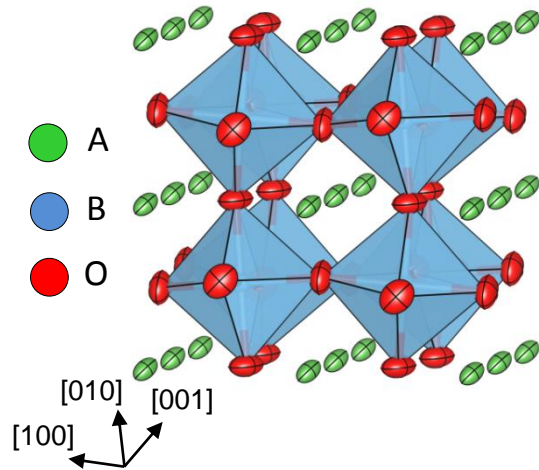


Phonon localisation in $\text{La}_{0.5}\text{Na}_{0.5}\text{TiO}_3$

Phonon localisation:

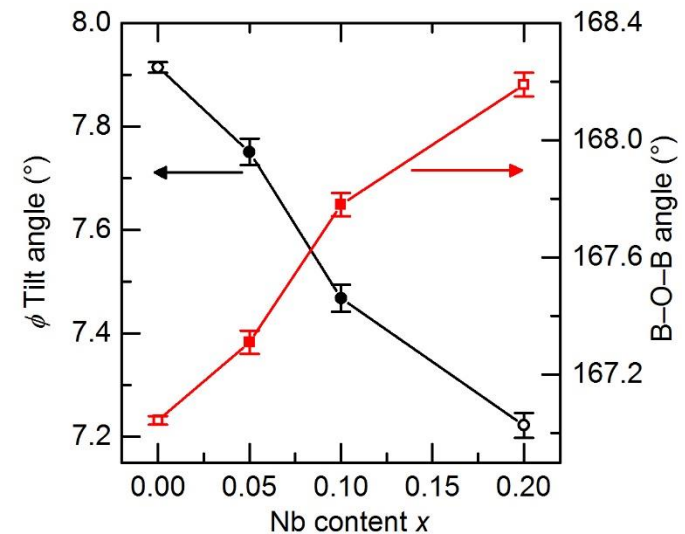
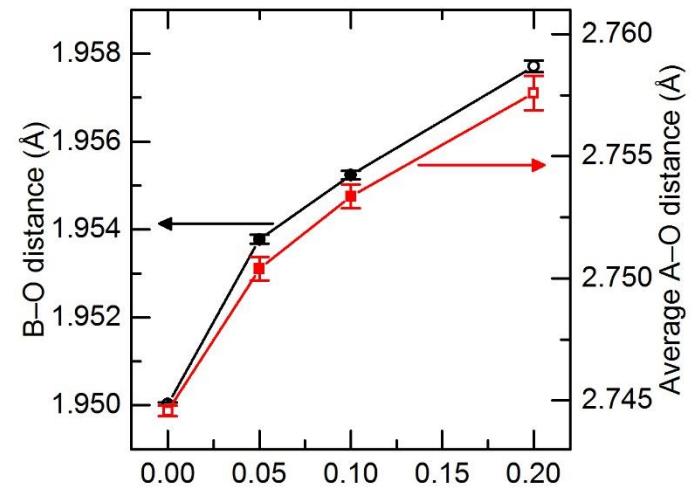
- Computed average separation (d_{average}) between 10 atoms exhibiting largest displacements for each phonon mode
- Coherent phonon transport in STO through well-established periodic nearest neighbour couplings
- Phonons are more localised in LNTO as a result of disorder



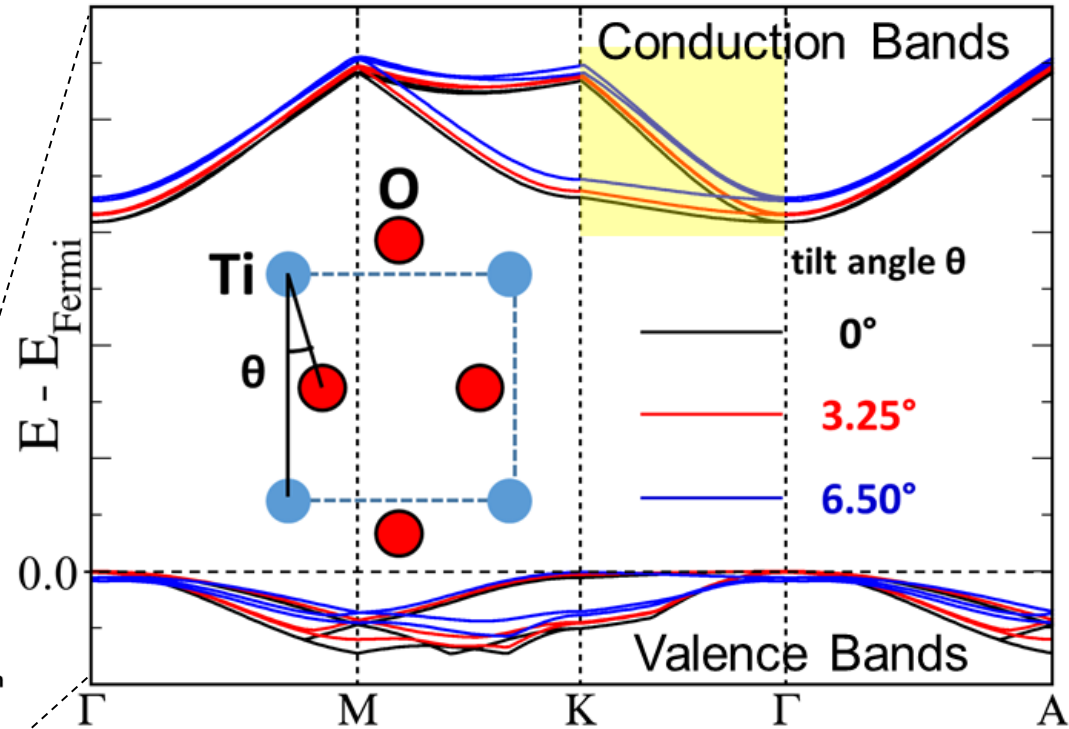
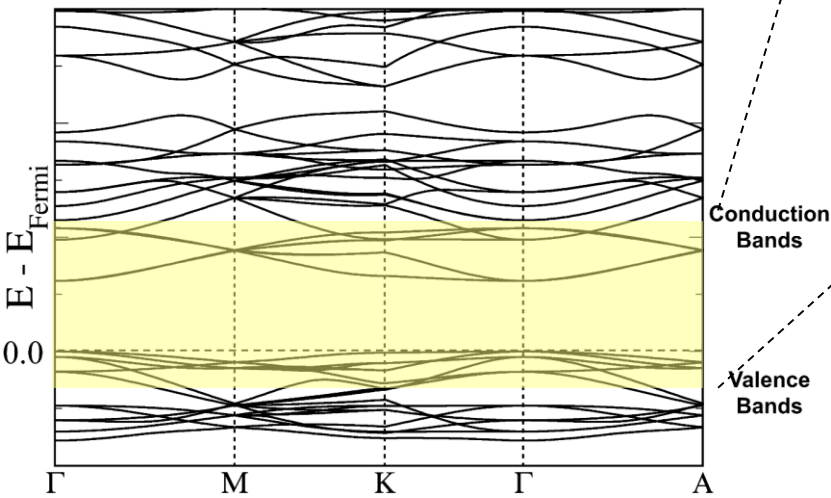
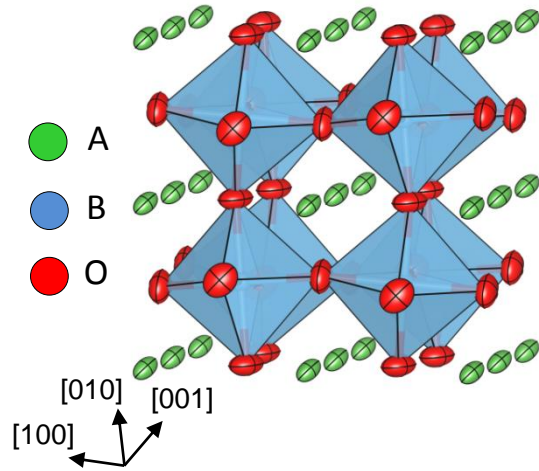


B site substitution:

- LNTO is hexagonal with $R\text{-}3c$ symmetry
- Substitution of Nb^{5+} expands unit cell and reduces octahedral tilting

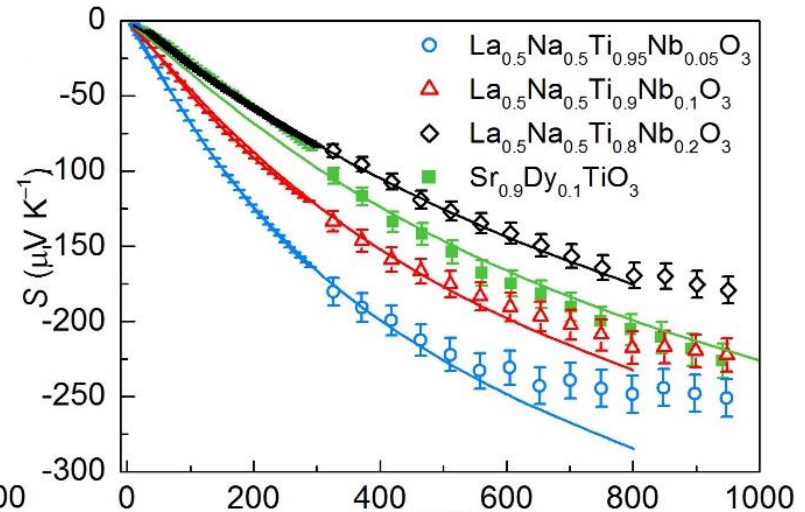
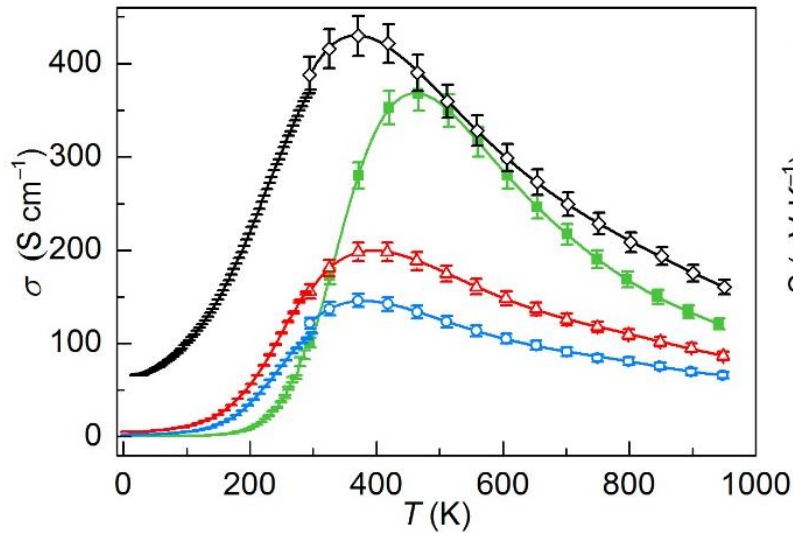


Band structures



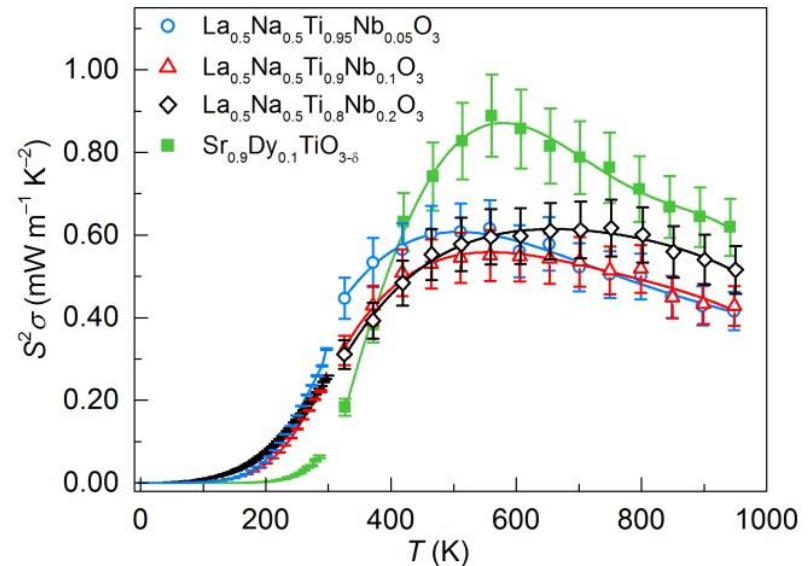
- Coexistence of dispersive and dispersion-less bands at the bottom of the conduction band is retained in LNTiO₃

Electronic properties

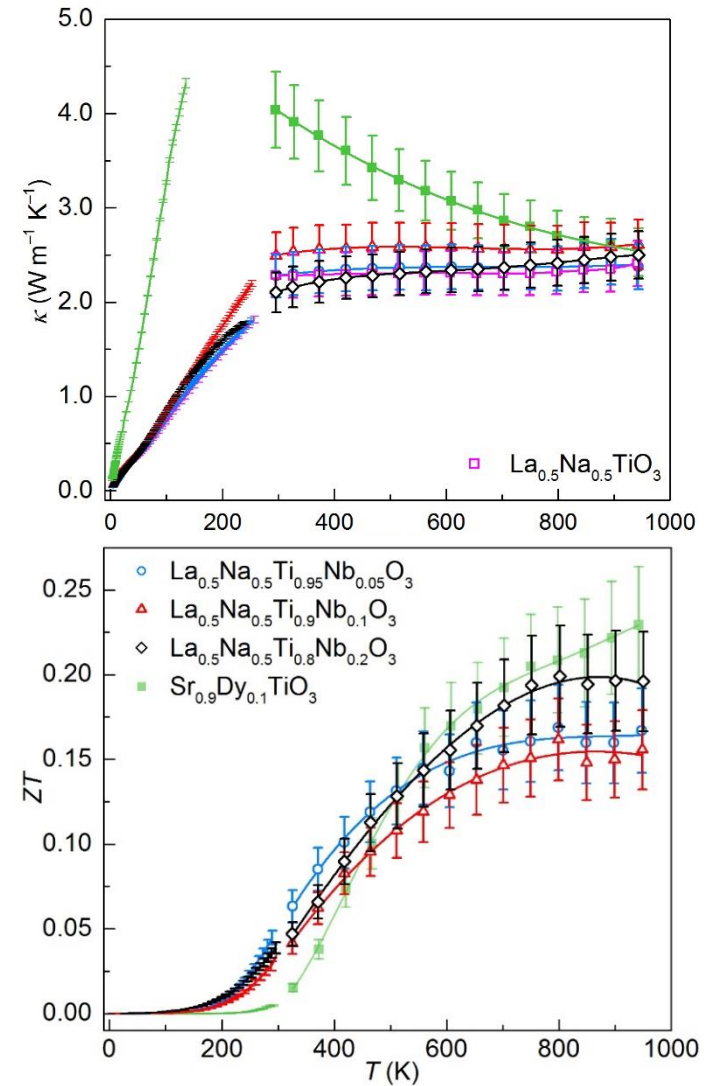
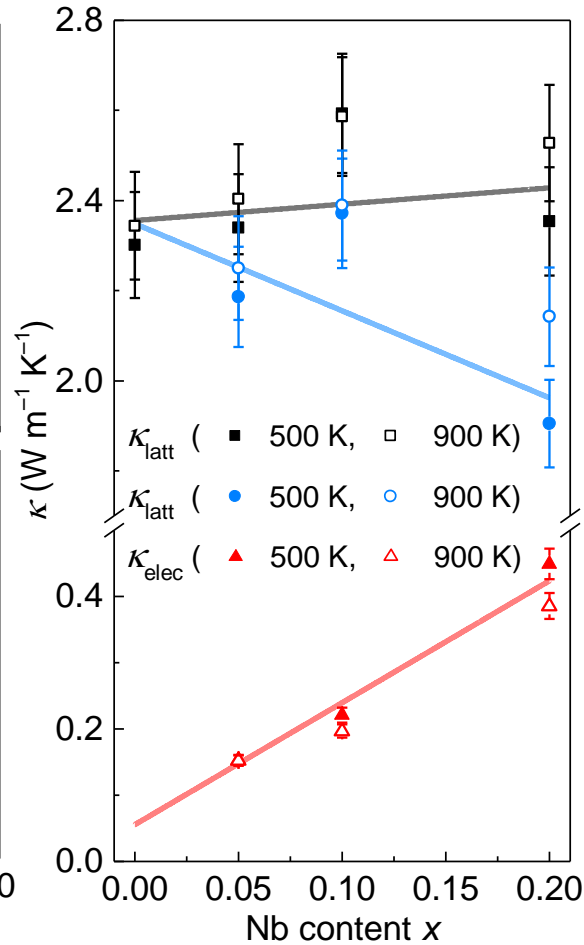
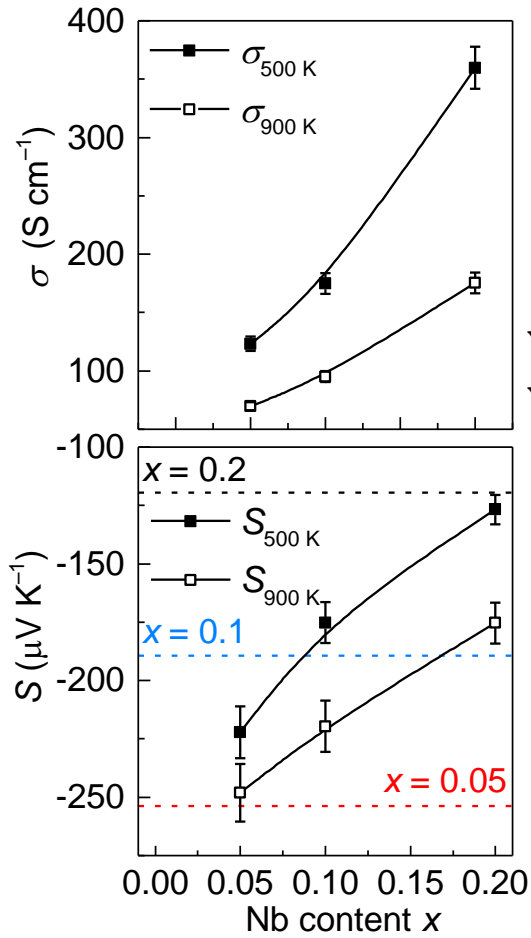


B site substitution:

- Increasing carrier concentration increases σ , but decreases S
- Able to extract carrier concentrations through fitting of Seebeck data
- Large grain boundary contributions to σ limit the power factor



$La_{0.5}Na_{0.5}Ti_{1-x}Nb_xO_3$



Conclusions & plans

Conclusions:

- Phonon engineering of SrTiO_3 through substitution to $\text{La}_{0.5}\text{Na}_{0.5}\text{TiO}_3$ results in phonon-glass thermal conductivity
- Chemistry of $\text{La}_{0.5}\text{Na}_{0.5}\text{TiO}_3$ analogous to that of SrTiO_3 , meaning it can be doped either through A or B site
- Phonon-glass thermal conductivity is retained upon B site substitution with donor Nb^{5+} which enhances electronic transport
- Thermal and electronic transport are largely decoupled in $\text{La}_{0.5}\text{Na}_{0.5}\text{Ti}_{1-x}\text{Nb}_x\text{O}_3$, yielding PGEC characteristics

Future plans:

- Explore high-mass contrast combinations to target other phonon glass systems
- Optimise processing of titanate ceramics to enhance PF further

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